

## Structural Characterization of a Pentacene Monolayer on an Amorphous SiO<sub>2</sub> Substrate with Grazing Incidence X-ray Diffraction

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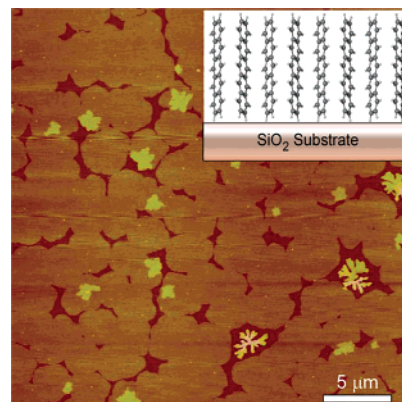
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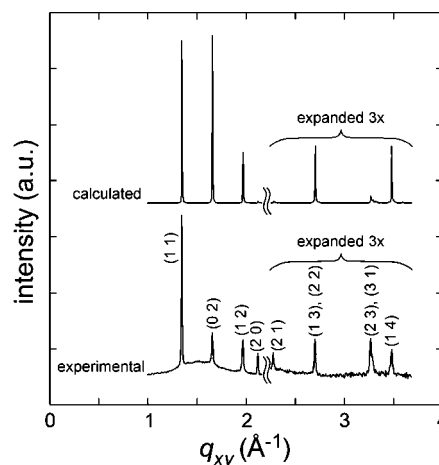
Pentacene, a simple linear oligoacene consisting of five fused benzene rings, has emerged as a viable candidate for the semiconducting transport layer in organic thin film transistors (OTFTs).<sup>1–4</sup> The interest in pentacene films, which exhibit p-type transport, primarily stems from their relatively high hole mobilities and high on-to-off current ratios in OTFTs.<sup>5</sup> Carrier transport in the channel between the source and drain electrodes is thought to occur in the first few layers of the semiconductor, or perhaps the first layer, in proximity with the dielectric layer adjacent to the gate electrode.<sup>6,7</sup> It is also widely recognized that the transport properties of crystalline organic films depend strongly on the intermolecular overlap of electronic wave functions within the semiconductor layer, which is very sensitive to the molecular packing in the crystal.<sup>8,9</sup> Surprisingly, little is known about the detailed crystal structure of the active transport layers in OTFTs, including pentacene films. We report herein preliminary grazing-angle incidence X-ray diffraction (GIXD) data for a monolayer-thick pentacene film grown on amorphous silicon dioxide (*a*-SiO<sub>2</sub>), a commonly used dielectric layer in OTFTs. The data confirm that the monolayer is crystalline and has a structure that differs from that of bulk pentacene, which has important implications for carrier transport in pentacene-based OTFTs.

The crystal structure of bulk pentacene consists of layers of pentacene molecules arranged in a herringbone packing motif with an interlayer spacing of  $d_{001} = 14.1 \text{ \AA}$ .<sup>10,11</sup> Three “thin film” multilayer phases with different  $d_{001}$  values of 14.4, 15.0, and 15.4 Å have been identified by wide-angle X-ray diffraction.<sup>12,13</sup> The selectivity toward these multilayer phases appears to be governed by a variety of factors including substrate material, substrate temperature during deposition, rate of deposition, and film thickness.<sup>12,14,15</sup> The different  $d_{001}$  values imply dissimilar packing of the pentacene molecules in the *ab* plane, which is regarded as the high-mobility plane for hole transport in pentacene-based OTFTs. Despite its importance to transport, structural characterization of the *ab* plane in the multilayer phases has been limited to substrates other than *a*-SiO<sub>2</sub>,<sup>12,13,16–18</sup> and the crystal structure of a pentacene monolayer on any substrate, including *a*-SiO<sub>2</sub>, has not been reported.

Atomic force microscopy (AFM) of a vacuum sublimed pentacene monolayer<sup>19</sup> on *a*-SiO<sub>2</sub> reveals micrometer-sized domains with a thickness of  $16.0 \pm 0.6 \text{ \AA}$  (Figure 1).<sup>20</sup> Characterization of this pentacene monolayer by GIXD at room temperature<sup>21</sup> afforded a diffraction pattern (Figure 2) that could be indexed to a near-rectangular in-plane unit cell with dimensions  $a = 5.916 \text{ \AA}$ ,  $b = 7.588 \text{ \AA}$ , and  $\gamma = 89.95^\circ$ . These values differ from the corresponding lattice parameters reported for bulk pentacene ( $a = 6.266 \text{ \AA}$ ,  $b = 7.775 \text{ \AA}$ , and  $\gamma = 84.684^\circ$ ), but are consistent with a packing



**Figure 1.** A tapping-mode AFM image, acquired in air under ambient conditions, of monolayer-thick pentacene domains on a 3000 Å thick *a*-SiO<sub>2</sub> film. Inset: Pentacene monolayer structure based on GIXD film.



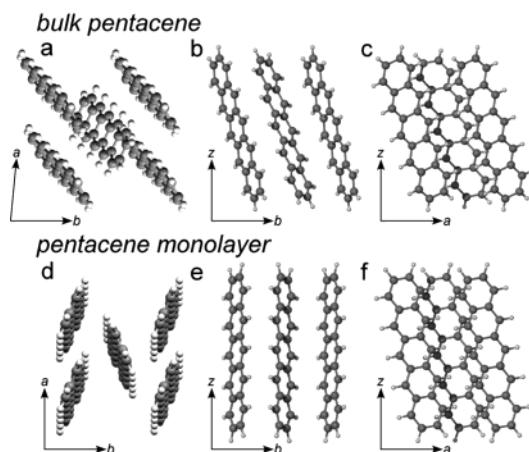
**Figure 2.** GIXD pattern (bottom) for a pentacene monolayer and a diffraction pattern (top) calculated for an energy-minimized crystal structure model based on the GIXD lattice parameters and the (001) layer motif of bulk pentacene as the starting point.

motif resembling the (001) layers in the bulk form, that is, with the pentacene molecules in the monolayer adopting a near-vertical orientation on the *a*-SiO<sub>2</sub> substrate. The diffraction peaks were very narrow, consistent with crystallite sizes >70 nm and in agreement with the large domains observed by AFM. The peak widths increase with increasing  $q_{xy}$  beyond that expected from Debye–Scherrer broadening, suggesting small variations in the in-plane  $d$ -spacings of the crystallites.<sup>22</sup> A broad feature centered at  $q_{xy} = 1.55 \text{ \AA}^{-1}$  is due to the amorphous SiO<sub>2</sub> substrate.

A model of the monolayer structure was constructed using the room-temperature single-crystal structure of bulk pentacene as a

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**Figure 3.** Normal views of the  $ab$  planes of bulk pentacene and the model monolayer structures (left) and the respective side views (center, right). The  $z$ -axis is the normal to the  $ab$  plane.

starting point. The  $a$  and  $b$  lattice parameters of the bulk structure were adjusted to the monolayer values, and the interlayer  $d_{001}$  spacing was expanded to an arbitrarily large distance (800 Å) to mimic an isolated monolayer. The lowest energy monolayer structure (ignoring contributions from the substrate) was then determined using the universal force field within the Cerius<sup>2</sup> environment (Accelrys).<sup>23</sup> The diffraction pattern produced by this energy-minimized monolayer structure is in reasonable agreement with the GIXD data, although the integrated intensities for some of the diffraction peaks for the preliminary model structure differ from those observed experimentally, particularly the relative intensities of the (11) and (02) peaks. Some of the discrepancy can be attributed to contributions from the  $a$ -SiO<sub>2</sub> background, particularly in the lower  $q_x$  region. We anticipate that a more complete data set, particularly data collected at  $q_z > 0$ , will permit precise determination of the pentacene tilt and an improved refinement of the crystal structure.<sup>24</sup> Nevertheless, the GIXD data demonstrate unequivocally that the pentacene monolayer on the  $a$ -SiO<sub>2</sub> dielectric layer is highly crystalline and has a structure that is distinguishable from the bulk.

The energy-minimized monolayer structure exhibits a herringbone packing of the pentacene molecules similar to that observed in the bulk (Figure 3). The herringbone “edge-to-face” angle between pentacene molecules is 48.1°, as compared to 52.3° angle in bulk pentacene. The pentacene molecules in the bulk phase tilt 25.1° with respect to the surface normal ( $z$ ) along an azimuthal angle of 139.7° (clockwise) with respect to the  $a$  axis. In contrast, the pentacene molecules in the model structure are tilted along the  $a$  axis by 11° with respect to  $z$ , and the tilt along the  $b$  axis is negligible. On the basis of this tilt and an estimated length of ca. 16.4 Å for the pentacene molecule, the anticipated monolayer thickness is 16.1 Å, in good agreement with the thickness measured by AFM.

These results demonstrate that a pentacene monolayer film on  $a$ -SiO<sub>2</sub> – the dielectric layer often used in OTFTs – is highly crystalline and has a structure that differs from the (001) layers in single crystals of bulk pentacene. The structural parameters of the monolayer also differ from most previously reported multilayers, although they are near those reported recently for pentacene multilayer films on a (100) NaCl substrate (by TEM).<sup>18</sup> GIXD is

uniquely suited for unraveling the structure–property relationships associated with carrier transport in these films because it can probe the region near the dielectric layer where transport most likely occurs. Furthermore, GIXD can be used to characterize films on various substrates and during application of gate bias, providing much-needed insight into the factors governing performance of OTFTs under actual operating conditions.

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